



Visible light active Fe doped TiO₂ nanowires grown on graphene using supercritical CO₂

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ARTICLE INFO

Article history:

Received 2 June 2011

Received in revised form 11 August 2011

Accepted 12 August 2011

Available online 2 September 2011

Keywords:

Fe doped TiO₂/graphene sheets

Super critical carbon dioxide

Sol–gel method

Band gap reduction

Electron hole recombination

Photocatalyst

ABSTRACT

Visible light driven TiO₂ photocatalyst nanowire arrays doped with Fe were grown on the surface of functionalized graphene sheets (FGSSs) using a sol–gel method in the green solvent, supercritical carbon dioxide (scCO₂). The morphology of the synthesized catalysts was studied by SEM and TEM, which showed uniform formation of Fe doped TiO₂ nanowires on the surface of the graphene sheets, which acted as a template for nanowire growth through surface –COOH functionalities. Increasing Fe content in the nanowires gave only small changes in morphology but significantly higher BET surface areas. Optical properties of the synthesized composites were examined by UV and PL spectroscopy which showed a significant reduction in band gap with increasing Fe content, i.e. decreasing from 3.2 to 2.3 eV at 0.6% Fe. High resolution XPS and Raman analysis showed the interaction of Fe with the TiO₂ lattice and also bonding of TiO₂ with –COOH groups on the surface of the graphene sheets. The photocatalytic properties of the prepared catalysts were evaluated under visible light solar irradiation for the photodegradation of 17 β -estradiol (E2), an endocrine disrupting hormone which is commonly released into aquatic environments. All prepared catalysts with different ratios of Fe were active in the visible region of the solar spectrum with the photocatalytic activity significantly enhanced with increasing Fe doping levels with a plateau at 0.6–0.8% Fe. The prepared catalysts showed higher activity than both Fe doped TiO₂ and TiO₂/FGSSs composites.

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1. Introduction

Heterogeneous photocatalysis has attracted considerable attention in recent years due to the potential for removing emerging environmental contaminants such as endocrine disrupting hormones (EDHs) [1–3]. TiO₂-based photocatalysts have several advantages owing to titania's abundance, non-toxicity, low-cost, unique optical-electronic properties and long term thermodynamic stability [4]. Nanostructured TiO₂ has been used for various environmental applications including degrading organic pollutants [5], dye-sensitized solar cells [6], self cleaning surfaces [7], water splitting [8] and antimicrobial coatings [9,10]. However, there are limitations for using commercial nano TiO₂ in these applications (e.g. Degussa P25) as TiO₂ nanoparticles easily agglomerate, have low surface areas, and give a low efficiency because of TiO₂'s wide band gap (3.2 eV) which is only active in the UV region of the solar spectrum [11]. Current research efforts are focussed on enhancing photocatalytic degradation using visible light (400 nm < λ < 760 nm) which is often unused in

UV-photocatalysis, although a target for next-generation photocatalytic processes.

For extending the absorbance of TiO₂ into the visible region, doping with transition metals is a useful technique [12]. For doping of TiO₂, various metal ions have been used, but among them, Fe³⁺ is considered as a strong candidate as it has a similar radius to Ti⁴⁺ and can easily fit into the crystal lattice of TiO₂ [12–14]. Moreover, the redox potential (energy differential) of Fe²⁺/Fe³⁺ is close to that of Ti³⁺/Ti⁴⁺, resulting in shifting its optical absorption into the visible region [12,14]. Another common method for increasing the photocatalytic efficiency of TiO₂ is immobilization of TiO₂ powder on a co-adsorbent surface such as zeolites [15], alumina [16], silica [17] or activated carbon [18]. Among these, carbonaceous materials are of tremendous current interest due to their unique pore structure, electronic properties, adsorption capacity and acidity. These materials include activated carbon [19], carbon nanotubes, and more recently graphene [20–23]. Graphene can be synthesized inexpensively from earth abundant graphite [24] providing larger specific surface areas similar to activated carbon and carbon nanotubes. This facilitates pollutant adsorption enhancing the photocatalytic efficiency by TiO₂. In addition, functional groups such as alcohol and carboxylic acids can be introduced onto the graphene surface [25]. Groups such as –COOH allow for both

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adsorption of specific pollutants and potentially can be exploited as a selective template for sol–gel nanowire growth on the graphene surface. Hence, graphene can potentially harness and stabilize different semiconductors on its surface to produce more stable and porous photocatalysts [26,27].

There are a few reports for the preparation of graphene-TiO₂ composites and also their corresponding applications. P25-graphene composites synthesized by a hydrothermal method were used as a photocatalyst for photodegradation of methylene blue. A remarkable enhancement in reaction rate was observed compared to P25 and P25-CNTs [22]. Titanium oxide nanoparticle-graphene oxide (TiO₂-GO) and titanium oxide nanoparticle-reduced graphene oxide (TiO₂-RGO) composites were synthesized by Lambert et al. via the hydrolysis of TiF₄ at 60 °C [28].

In this work, we show that Fe doped TiO₂ can be grown on the surface of functionalized graphene sheets (FGSs) containing –COOH functional groups using supercritical carbon dioxide (scCO₂) as the enabling solvent. TiO₂ nanowires were prepared previously in scCO₂ using Ti alkoxides and acetic acid as the polycondensation agent with the mechanism of wire formation examined [29]. ScCO₂ is an inexpensive, green alternative to conventional organic solvents [29,30] which is inexpensive, environmentally benign and non-flammable with low viscosity, zero surface tension and high diffusivity which are favourable for synthesizing fine and uniform nanomaterials. ScCO₂ can facilitate doping in titania [31,32] and has been used to enhance exfoliation of graphene sheets [33]. No drying process is required as complete removal of excess acetic reaction ingredients is easy by supercritical fluid extraction, maintaining the porous nanostructure [29,30]. The resulting materials from this study are examined with visible light for the photocatalytic degradation of 17β-estradiol (E2).

2. Experimental

2.1. Materials

Graphite flake nominally sized at 7–10 μm was provided from Alfa Aesar. Fuming nitric acid (>90%), sulphuric acid (95–98%), potassium chlorate (98%), titanium isopropoxide (99.999%) trace metals basis, iron chloride, FeCl₃, titanium (IV) oxide nanopowder (99.7%) trace metals basis (anatase), E2 (C₁₈H₂₄O₂, CAS registry number: 50-28-2) and hydrochloric acid (37%) were purchased from Sigma–Aldrich and used as received.

2.2. Methods

2.2.1. Preparation of functionalized graphene sheets (FGSs)

Graphite oxide was prepared using the Staudenmaier method [34] with graphite (5 g) reacted with concentrated nitric acid (45 mL) and sulphuric acid (90 mL) with potassium chlorate (55 g). Potassium chlorate was added slowly over 15 min to avoid sudden increases in temperature. The mixture was stirred at room temperature for more than 3 days. After reaction completion, the viscous dark green mixture was added to excess water, washed with a 5% solution of HCl, and then repeatedly washed with water until the pH of the filtrate was neutral. The prepared graphite oxide was kept in a vacuum oven at 80 °C until use. Then, the graphite oxide powder was placed in a quartz tube with an inlet and outlet for continuously passing nitrogen gas. The tube was then inserted into a tubular furnace preheated to 1050 °C and kept at this temperature for 30 s, and then removed. The graphite oxide was reduced and expanded to form FGSs.

2.2.2. Preparation of Fe doped TiO₂/graphene sheets composites

TiO₂ nanowires were successfully synthesized on the surface of graphene sheets in supercritical carbon dioxide (scCO₂) using

acetic acid/titanium isopropoxide (AcOH/TIP: 4), 60 °C and 5000 psi as previously determined as the optimized experimental conditions [35]. In a typical experiment, first FGSs (0.05 g) were placed in a 10 mL high pressure view cell followed by quick addition of titanium isopropoxide (1 mL), FeCl₃ dissolved in isopropanol (Fe percentage is the weight ratio of FeCl₃:TIP), acetic acid (4 mL) and CO₂ to the desired pressure and temperature. A magnetic stirrer with TFE stir bar was used for mixing the reaction mixture. It was found that the mixture of iron chloride/isopropanol, functionalized graphene sheets, titanium isopropoxide and acetic acid were miscible with CO₂ at 60 °C and 5000 psig pressure. The stirrer was stopped after 24 h while normally several days of aging were required for complete reaction. After aging, the formed gel was washed continuously using 80 mL of CO₂ at a rate of approximately 0.5 mL/min, followed by controlled venting at 0.5 mL/min to prevent collapse of the solid network. The resulting powder was then calcined at 450 °C in air (tubular furnace) using a heating rate of 10 °C/min for 2 h. The calcined powder was kept in a vacuum oven at 80 °C.

2.3. Characterization

The nanostructured morphologies of the samples were examined using scanning electron microscopy (SEM) micrographs (Model LEO 1530) and transmission electron microscopy (TEM) (Model JEOL 2010F) with a 200 keV Schottky field emission microscope (TEM/STEM). Samples for SEM imaging were prepared by applying the powder directly to a carbon adhesive tape. For TEM analysis, the powdered samples were dispersed in methanol by sonication and then placed on a copper grid covered with holey carbon film and dried by normal evaporation. Structural analysis of the samples was performed using an X-ray powder diffractometer (Rigaku Miniflex XRD, Texas, USA), fitted with a rotating sample holder, a scintillation counter detector and a divergent beam utilizing a Cu Kα source of X-rays (λ = 1.5418 Å). Raman analysis was performed using a Kaiser optical systems (RXN1-785) with 5 times exposure and 20 times accumulation. The XPS analyses were carried out with a Kratos Axis Ultra spectrometer using a monochromatic Al Kα source (15 mA, 14 kV). Photoluminescence (PL) spectra were carried out using a PTI Quantamaster 50 spectrophotometer with the Xenon lamp (75 V). UV–Vis spectra were performed with a Shimadzu 3600 deuterium arc lamp and two different detectors (PMT and PbS). An integrating sphere was utilized in order to measure the diffuse reflectance of the material.

2.4. Photocatalytic activity measurements

Photodegradation of 17β-estradiol (E2) as a model endocrine disrupting compound (EDC) was investigated under visible solar irradiation (λ ≥ 420 nm). A solar simulator (Model: SS1KW, Sciencetech) with a 1000 watt xenon arc lamp equipped with an air mass filter (AM 1.5G) with a special UV cut-off filter (290 < λ < 420 nm) provided the visible light source. Typically, a mixture of aqueous E2 solution (5 μg/L) and 0.5 g/L of catalyst was vigorously stirred for 30 min to establish an adsorption/desorption equilibrium in the dark. Then the reaction solution was irradiated under visible light in an open water-jacketed vertical photo-reactor which was placed on a magnetic stirrer during all experiments, under aerated conditions [1]. The temperature of the reactions was controlled at 22 ± 1 °C using a controlled temperature bath. At given time intervals, 5 mL aliquots were sampled and centrifuged to remove the particles. The filtrates were analyzed by HPLC (ICS 300, Dionex), using a PDA UV detector, connected to Chromeleon software. Separations were carried out with an Acclaim 120 C18 reversed-phase column (150 mm × 4.6 mm i.d., 5 μm particle size, Dionex, USA). The injection volume was 40 μL from a 2 mL HPLC

Table 1Synthesis conditions, morphology, BET surface area, pore volume, and pore size distribution of TiO₂, Fe doped TiO₂/FGSs composites (0.1%, 0.2%, 0.4%, 0.6% and 0.8%).

Materials	BET surface area (m ² /g)	Total pore volume (cm ³ /g)	Average pore size (nm)	Band gap (eV)	Morphology
TiO ₂ anatase (commercial)	50	–	–	3.2	Nanoparticles
TiO ₂ /FGSs	118	0.12	45.3	2.9	Nanowires
Fe/TiO ₂ (0.4%)	156	0.11	38.7	2.5	Nanoflower
Fe/TiO ₂ /FGSs (0.1%)	128	0.10	41.6	2.82	Nanowires on the sheet
Fe/TiO ₂ /FGSs (0.2%)	174	0.13	39.3	2.53	Nanowires on the sheet
Fe/TiO ₂ /FGSs (0.4%)	214	0.20	37.48	2.42	Nanowires on the sheet
Fe/TiO ₂ /FGSs (0.6%)	256	0.23	31.5	2.25	Nanowires on the sheet
Fe/TiO ₂ /FGSs (0.8%)	297	0.29	29.2	2.25	Nanowires on the sheet

All materials prepared in scCO₂ (60 °C, 5000 psi), FGSs/TiP: 1:20, AcOH/TiP: 4 and calcined at 450 °C.

vial, capped and sealed with PTFE lid. The mobile phase was a mixture of AcN and Milli-Q water (50:50, v/v) at a flow rate of 1 mL/min. The column temperature was maintained at 30 °C and detection wavelength was set at 280 nm, the maximum absorbance of E2. The retention time of E2 in the HPLC column was 5.33 min. In the kinetic degradation experiments of E2, the initial concentration was $5 \pm 0.3082 \mu\text{g/L}$. The maximum degradation was kept to 90%, i.e. the final concentration of E2 measured was approximately $0.5 \mu\text{g/L}$, which was within the limits of the determined calibration curve (see [Supporting Document](#)).

3. Results and discussion

3.1. Characterization of synthesized materials

Different percentages of Fe doped TiO₂/functionalized graphene sheets (FGSs) and Fe doped TiO₂ were synthesized in scCO₂. The experimental conditions and resulting properties of the synthesized materials are summarized in [Table 1](#).

The morphology and particle size of the Fe doped TiO₂/FGSs materials synthesized via the examined acid-modified sol–gel technique in scCO₂ were assessed by SEM and TEM analysis.

In [Fig. 1](#), a typical SEM image of Fe doped TiO₂/FGSs (0.2% Fe – [Fig. 1a](#)) and (0.6%Fe – [Fig. 1b](#)) is shown. Increasing the amount of Fe was not found to change the morphology of nanowires on the surface of the graphene sheets significantly; however, the nanowires with higher Fe contents appear more porous. As shown in [Table 1](#), the BET determined surface areas of the Fe doped TiO₂/FGSs samples were found to be significantly higher than the undoped TiO₂ nanowires on the graphene sheets, and directly increased with increasing Fe content. Introducing Fe to the titania sol–gel mixture for growth on the graphene sheets increases the surface area by producing more pores inside the composites with smaller pore sizes enhancing the mesoporous structure. Lucky and Charpentier synthesized Fe doped TiO₂ nanofibers in scCO₂ without graphene, reporting similar results [[31](#)].

In [Fig. 1c](#) and [d](#), TEM images show that the diameter of the TiO₂ nanowires is less than 20 nm. Good dispersion of TiO₂ nanowires on the graphene sheets can be observed along with

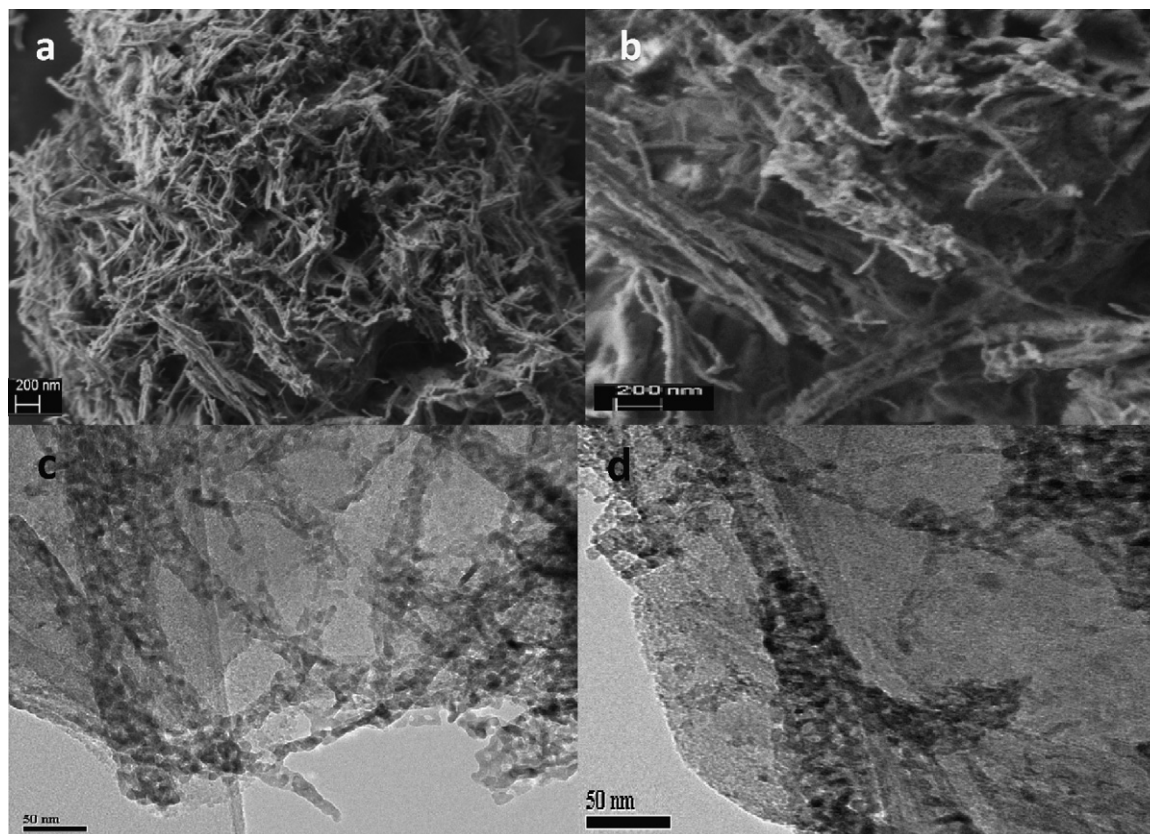


Fig. 1. SEM images of (a) Fe doped TiO₂/FGSs 0.2% (200 nm scale), (b) 0.6% (200 nm scale); TEM images of Fe doped TiO₂/FGSs (c) 0.2% (50 nm) and (d) 0.6% (50 nm).

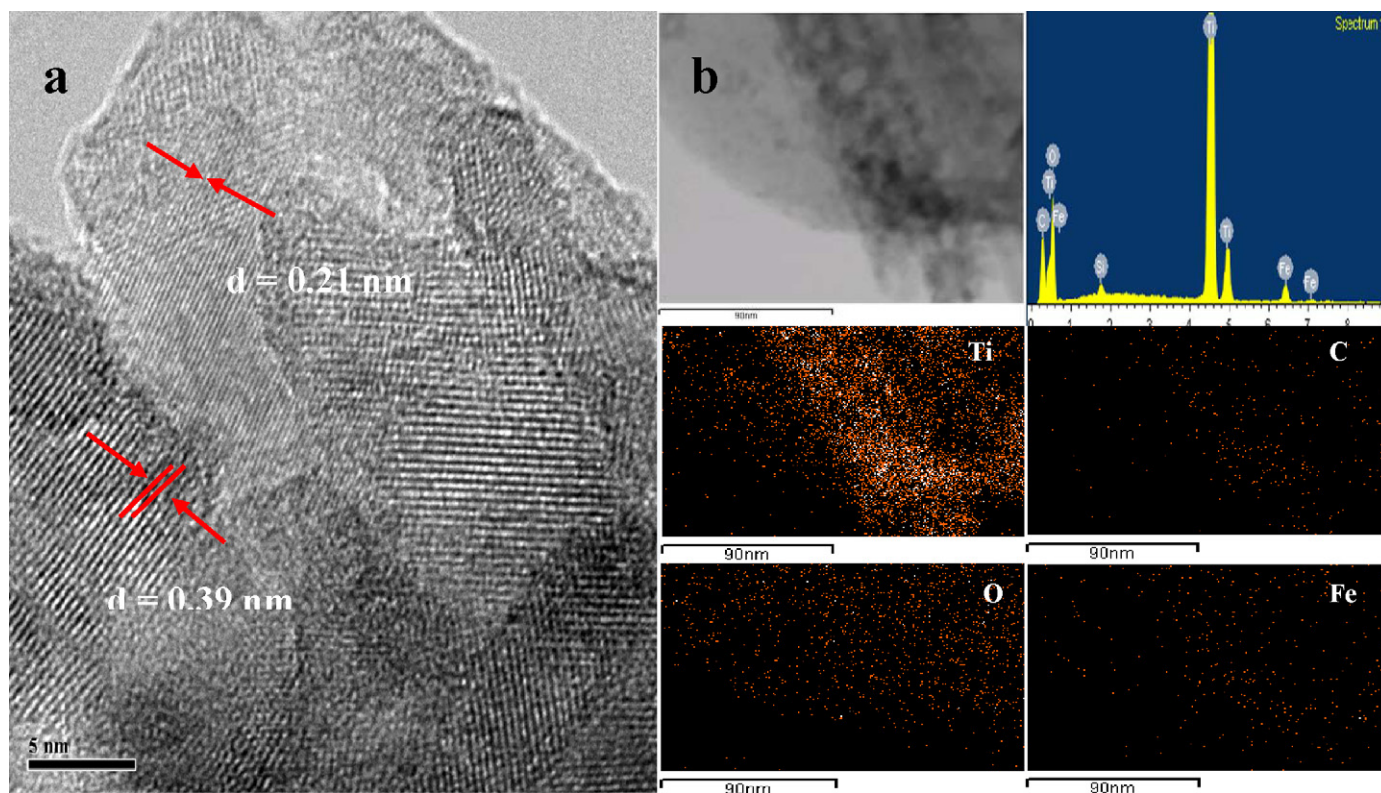


Fig. 2. (a) HRTEM image of Fe doped TiO₂/FGSs (0.2%), (b) HRTEM-EDS elemental mapping of Fe doped TiO₂/FGSs assembly.

excellent contact between the graphene sheet and wires. SEM images of 0.2% Fe doped TiO₂ and TiO₂/FGSs are available in the [supporting information](#) for comparison. When graphene sheets were not used as the support, the morphology of synthesized materials showed a flower-like structure. TiO₂/FGSs composites also showed similar structure as Fe doped TiO₂/FGSs with less porosity.

The HRTEM micrographs of the lattice image of 0.2% Fe doped titania/graphene calcined at 450 °C is given in Fig. 2a, showing uniform dispersion of anatase with *d*-spacing 0.39 nm (1 0 1) as well as some rutile with *d*-spacing 0.21 nm (1 1 1) [36]. There was no Fe detected in the HRTEM micrographs indicating that all the Fe is contained within the TiO₂ lattice. However, the *d*-spacing increased which is in good agreement with the XRD data as explained later confirming expansion of the TiO₂ lattice. In Fig. 2b, the EDX spectra with elemental analysis of this sample shows 25.6% of C, 59.1% of O, 14.1% of Ti and 1.2% of Fe on the surface. Elemental mapping was measured in several selected sample areas confirming uniform dispersion of Ti, C, O and Fe in the synthesized material.

The optical absorption spectra obtained by the diffuse reflection method for TiO₂ nanowires and different percentages of Fe doped TiO₂ on the surface of graphene sheets prepared in scCO₂ are shown in Fig. 3a. By adding Fe to the TiO₂/FGSs composites, a noticeable shift in the absorption edge into the visible light region was observed (to 550 nm). As shown in the figure, the band gap energy of doped TiO₂ was obtained by drawing a tangent line on the edge of the UV spectra to determine the cut off wavelength at $y = 0$ [37,38]. Pure TiO₂ anatase has a band gap of 3.2 eV with the band gap decreasing with an increasing amount of Fe in the composites, with the smallest band gap measured of 2.25 eV for 0.6% Fe doping. Further increasing of the Fe concentration to 0.8% did not further decrease the band gap. In Fig. 3b, the diffuse reflectance UV–Vis

spectra of TiO₂ nanowires (1), 0.4% Fe doped TiO₂ (2) and 0.4% Fe doped TiO₂/FGSs (3) prepared in scCO₂ are compared. By adding graphene to the system, the band gap of Fe doped TiO₂ decreased from 2.5 to 2.4 eV attributed to enhanced electron hole recombination of TiO₂ and graphene. In Fig. 3c, the UV spectra of TiO₂ with band gap of 3.2 eV are compared to TiO₂/FGSs with a band gap of 2.9 eV.

The photoluminescence (PL) spectra of TiO₂ and the scCO₂ synthesized composites are shown in Fig. 4. TiO₂ (anatase) has a characteristic broad peak at around 540 nm as shown in spectra a [39]. Although the PL intensity decreased in both the Fe doped TiO₂ nanowires and the TiO₂ nanowires/FGSs composites compared to anatase alone, the reduction is significantly greater in the Fe doped TiO₂/FGSs composites. When Fe doped TiO₂ nanowires are in contact with the graphene sheets, excited electrons have a lower chance for recombination resulting in a decreased band gap and enhanced visible light activity. These results help to confirm the UV–Vis results.

The structure of Fe doped TiO₂/FGS composites that was synthesized using the sol–gel method in scCO₂ appear to be amorphous; however, by thermal annealing at 450 °C, depending on dopant loading, anatase or a mixture of anatase and rutile TiO₂ was obtained. In Fig. 5, XRD spectra of different percentages of Fe doped TiO₂/FGSs are shown. No obvious peaks for Fe could be detected indicating that the Fe ions were fully integrated into the TiO₂ crystal lattice as earlier observed by the HRTEM results. In the 0.2% Fe doped TiO₂/FGSs (Fig. 5a), a mixture of anatase and rutile TiO₂ is observed. By increasing the concentration of Fe to 0.4% (Fig. 5b), the amount of rutile phase decreased. By increasing the concentration of Fe further to 0.6% and 0.8% (Fig. 5c and d), only anatase phase was obtained. These results indicate that the specific concentration of Fe ions can affect the rate of anatase-rutile phase transformation. By increasing the amount of dopant in TiO₂, the oxygen vacancies

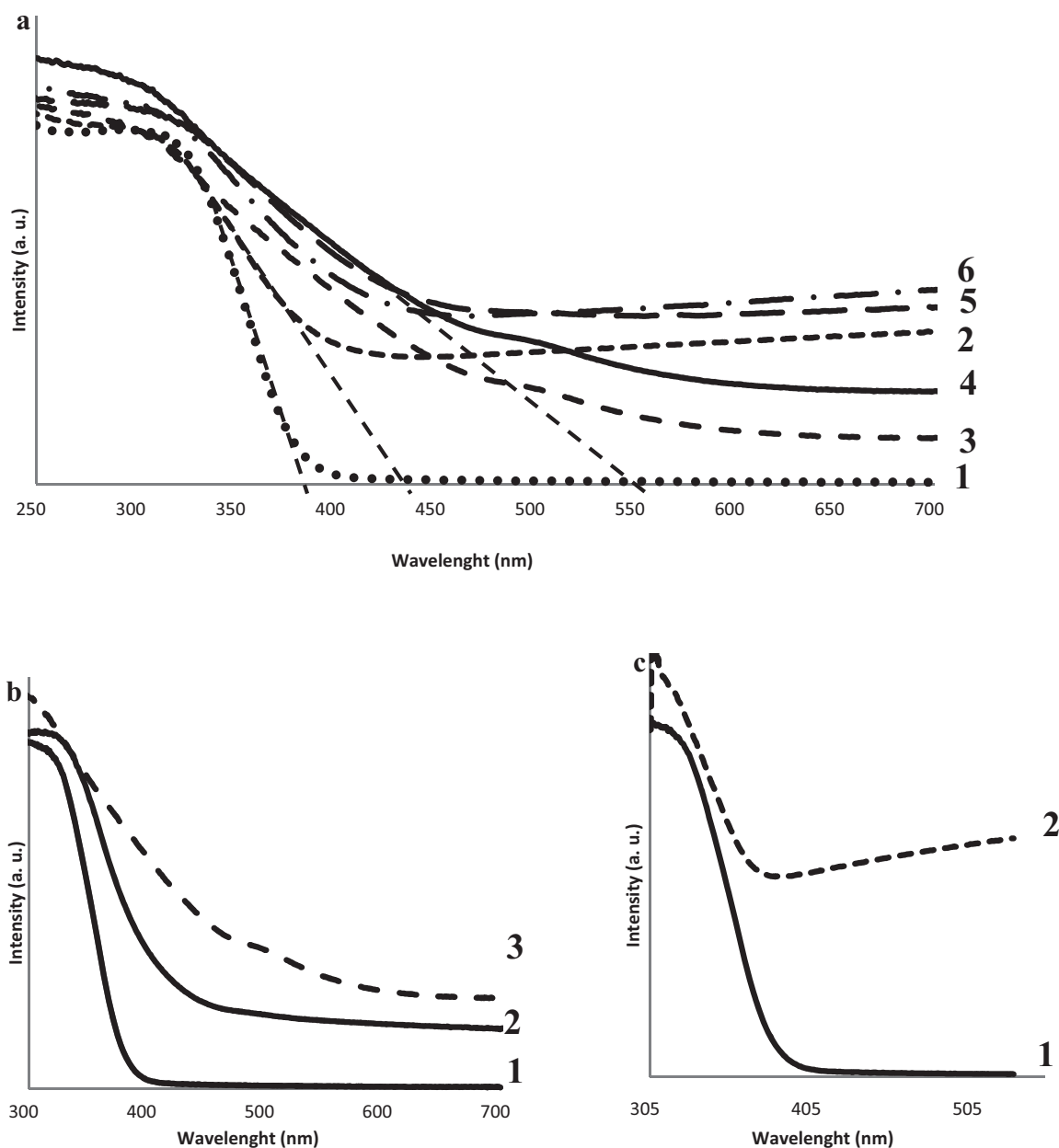


Fig. 3. Diffuse reflectance UV–Vis spectra of: (a) (1) TiO₂ anatase; Fe doped TiO₂/FGSS (2) 0.1%, (3) 0.2%, (4) 0.4%, (5) 0.6% (5) and (6) 0.8%; (b) (1) TiO₂, (2) 0.4% Fe doped TiO₂ and (3) 0.4% Fe doped TiO₂/FGSS; (c) (1) TiO₂ and (2) TiO₂/FGSS prepared in scCO₂.

increased and more anatase phase can result from A–R phase transformations involving a contraction of the oxygen structure [40]. Choi et al. studied the effect of different dopants on A–R phase transformation and suggested that the addition of dopants with small radii could be directly integrated into the TiO₂ crystal lattice and by introducing more oxygen sites in the lattice, more anatase TiO₂ was formed [41]. On the other hand, an increase in the Fe content leads to a shift of the (1 0 1) and (2 0 0) reflections to lower angles corresponding to the unit cell expansion due to the replacement of Ti⁴⁺ ions by slightly larger Fe³⁺ ions. The same changes in *d*-spacing with increasing Fe content were also observed in the HRTEM images of the doped materials.

Graphite has two well known characteristic bands in the Raman spectra; the strong G band at 1575 cm^{−1} and the weak D band at approximately 1355 cm^{−1} [42]. After oxidation of graphite to graphite oxide and then thermal expansion to FGSS, both the G and

D bands change noticeably. A higher disorder in graphite resulted in a broader G band, as well as a higher intensity, broader D band compared to that of the G band (Supporting Information) [42]. After functionalization of FGSS with TiO₂, the D/G ratio, which represents the degree of functionalization, increased dramatically indicating good attachment of the TiO₂ nanowires to the surface of the graphene sheets (D/G ratio ~2.5) as shown in Fig. 6. The main characteristic peak of anatase TiO₂ appears at 145 cm^{−1} which is attributed to the main *E_g* anatase vibration mode [42]. Moreover, the presence of crystalline TiO₂ in the composite is confirmed by the vibration peaks at 400 cm^{−1} (*B_{1g}*), 519 cm^{−1} (*A_{1g}*) and 640 cm^{−1} (*E_g*) [43]. After Fe doping, the Raman spectra did not change significantly (D/G ratio ~2.6) attributed to all the Fe being integrated into the TiO₂ lattice. The only change is a shift of the TiO₂ peak to the higher wavenumbers in the Fe doped composites, indicating an increase in the surface oxygen vacancies [44].

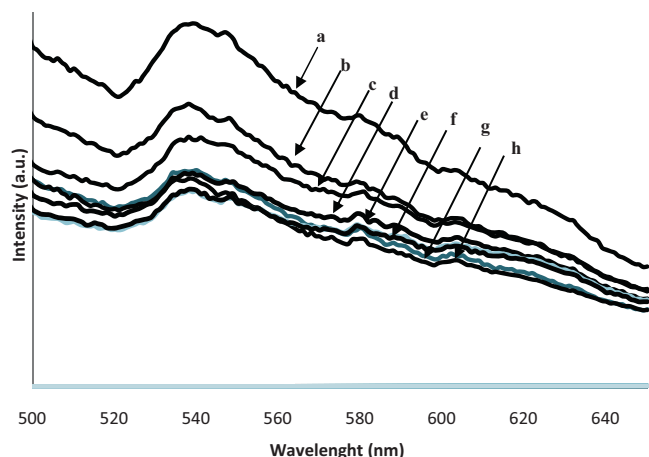


Fig. 4. Photoluminescence spectra of (a) TiO₂ (anatase), (b) TiO₂/FGSs composites, (c) Fe doped TiO₂, (d) 0.1%, (e) 0.2%, (f) 0.4%, (g) 0.6% and (h) 0.8% Fe doped TiO₂/FGSs composites.

Zhu et al. prepared Fe-doped nanocrystalline via a nonhydrolytic sol–gel method and observed a similar shift in the Raman study [45].

In the XPS spectra of FGSs, 90% carbon and 10% oxygen were detected. In high resolution XPS of C1s (see Supporting Document), four different peaks were observed at 284.7, 286.2, 287.7 and 289.3 eV which correspond to the (C–C, C=C), (C–H), (C–OH, C–O–C) and C=O functionalities, respectively. Akhavan studied the XPS spectra of graphite oxide and thermal expanded graphene sheets at different temperatures and reported similar results [46]. The XPS spectra of 0.6% Fe doped TiO₂/FGSs shows 6.1% Fe 2p, 54.7% O1s, 18.7% Ti 2p and 20.5% C1s. In high resolution of C1s, four different peaks of FGSs are still available but the intensity of peaks which represented C=O and C–OH decreased. The peak at 289.25 eV corresponds to C=O also shifted to lower energy (288.80 eV) implying coordination bonding between Ti and carboxylic acids on the surface of the graphene sheets. The O 1s main peak at 530.3 eV is assigned to the metallic oxides (Ti–O), which is consistent with the binding energy of O^{2–} in the TiO₂ lattices. The peak at 533.40 eV corresponding to O–H disappeared, but the peak at 531.38 eV

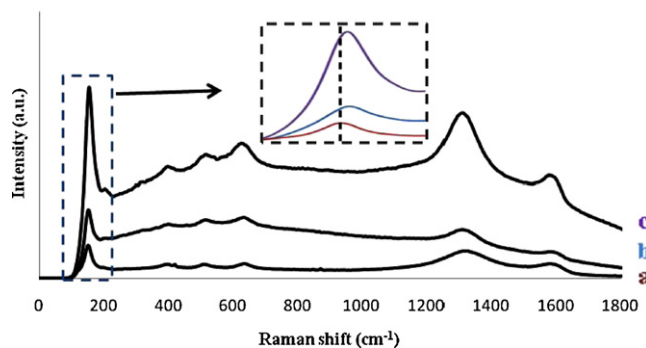


Fig. 6. Raman spectra of (a) TiO₂/FGSs composites, (b) Fe doped TiO₂/FGSs (0.2%) and (c) (0.6%).

corresponding to the carbonyl group is still observable [46]. The Ti 2p_{3/2} and Ti 2p_{1/2} for Fe doped TiO₂ on the graphene surface are located at binding energies of 458.36 eV and 464.1 eV, respectively, which is consistent with the values of Ti⁴⁺ in the TiO₂ lattices [14]. The peaks at 711.9 eV and 724.6 eV are assigned to 2p_{3/2} and 2p_{1/2} of Fe³⁺, respectively. These data exhibit a slightly positive shift compared to those in Fe₂O₃ (710.7 eV for 2p_{3/2} and 724.3 eV for 2p_{1/2}). The positive shift of the Fe2p level binding energy may be due to transferring the electrons from Fe³⁺ into the TiO₂ lattice and the formation of Fe–O–Ti bonds in the composite [14].

3.2. Visible-light photocatalytic activity

The photocatalytic activities of the different iron doped (0%, 0.2%, 0.4%, 0.6% and 0.8%) TiO₂ nanowire/graphene sheets and also Fe doped TiO₂ samples were evaluated through the photodegradation effects of E2 (Fig. 7) in aqueous solution using a solar simulator with a filter to block UV light allowing only visible irradiation ($\lambda > 420$ nm). We found no photodegradation of E2 in the presence of pure nano TiO₂ under visible irradiation as expected, as TiO₂ only produces OH[•] in the presence of UV radiation. The results for the different Fe doped TiO₂ nanowires/graphene sheets and also Fe doped TiO₂ samples are shown in Fig. 7. The degradation rate in visible light increased directly with increasing Fe dopant level, although only a marginal

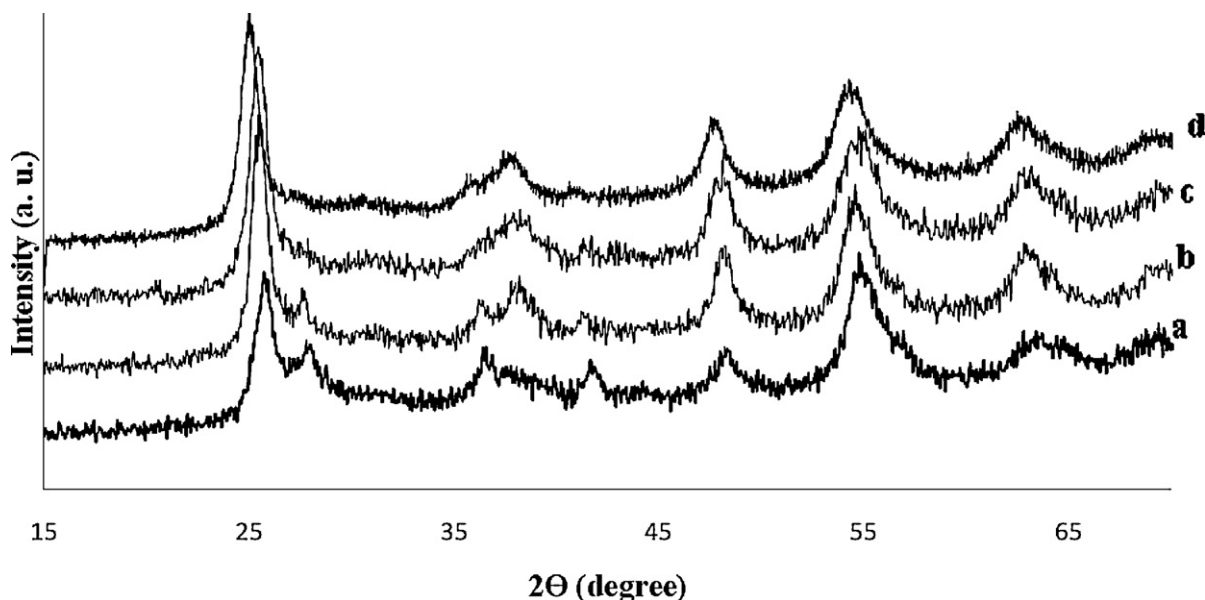


Fig. 5. XRD pattern of (a) 0.2%, (b) 0.4%, (c) 0.6% (c) and (d) 0.8% of Fe doped TiO₂/FGSs composites prepared at 60 °C and 5000 psi and calcined at 450 °C.

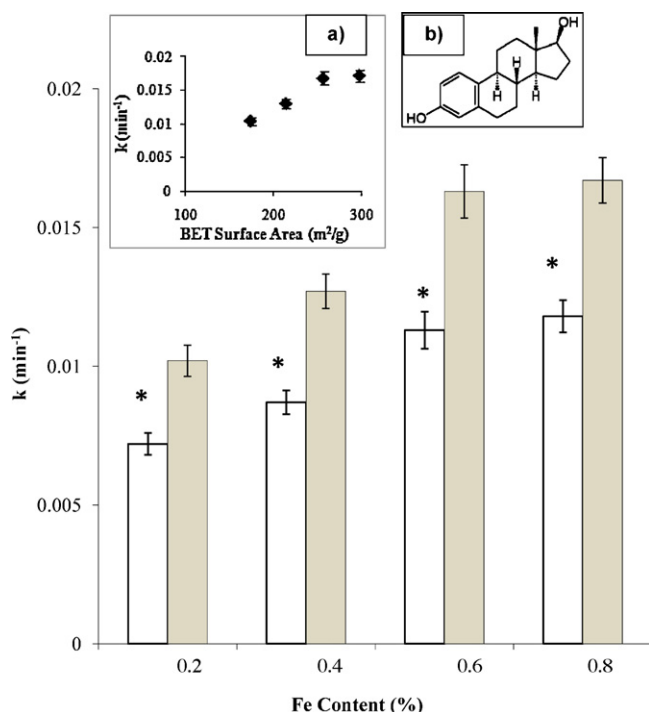


Fig. 7. Photocatalytic degradation rate constant of E2 using Fe doped TiO₂ samples on graphene sheets (*Fe/TiO₂ samples without graphene). Inset: (a) the effect BET surface area of Fe doped TiO₂ graphene sheets on the photocatalytic degradation rate constant of E2, and (b) structure of 17β-estradiol (E2).

increase in rate occurred from 0.6% to 0.8%, indicating the presence of an optimal loading. As shown in Fig. 7, the Fe doped TiO₂ nanowire/graphene assemblies show higher photocatalytic activity than the Fe doped TiO₂ composites without graphene, attributed to both enhancing the pollutants adsorption to the graphene surface along with graphene helping to lower titania's band gap energy.

The degradation kinetics are further studied at 0.6% Fe loading, as shown in Fig. 8. There was no photodegradation of E2 in the presence of visible light ($\lambda > 420$ nm) due to direct photolysis (i.e. no catalyst), because E2 does not have any absorbance in

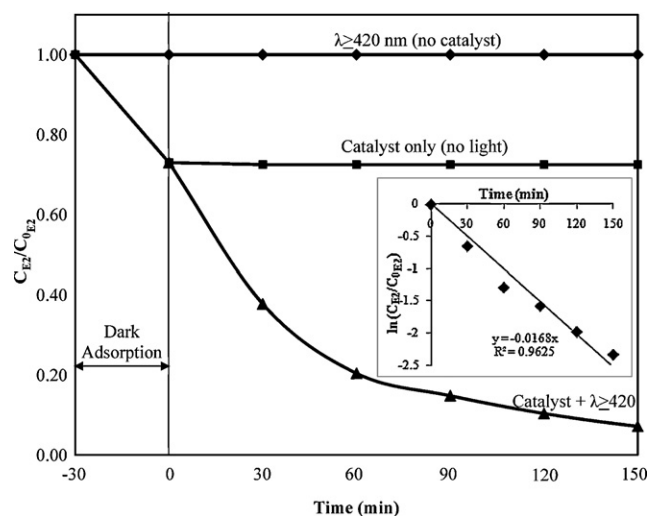


Fig. 8. Photocatalytic degradation and kinetic study of E2 under visible solar irradiation ($\lambda > 420$ nm). $C_{0E2} = 5 \mu\text{g/L}$, pH = 6.5, solar intensity = 1 SUN, catalyst: 0.6% Fe doped TiO₂ nanowires/graphene sheets and irradiation time = 150 min.

the visible wavelength region [1]. The experiments in the presence of photocatalyst illustrate that 30 min was sufficient to reach the adsorption equilibrium. The degradation of E2 could occur because of photocatalytic reactions in the presence of catalyst under visible light ($\lambda > 420$ nm), with E2 almost completely degraded in less than 150 min.

Reaction kinetics for adsorbing species are often described in terms of the Langmuir–Hinshelwood model, which can be expressed as:

$$-\frac{dC}{dt} = k_r \frac{k_a C}{1 + k_a C} \quad (1)$$

where $(-dC/dt)$ is the degradation rate of E2, C is the concentration of E2 in the solution, t is reaction time, k_r is the reaction rate constant, and k_a is the adsorption coefficient of the reactant. $k_a C$ is negligible when the value of C is very small. As a result, Eq. (2) can be described as pseudo-first order kinetics. Setting Eq. (2) at the initial conditions of the photocatalytic experiment, when $t = 0$, $C = C_0$, it can be described as:

$$\ln\left(\frac{C_0}{C}\right) = k_{app} \times t \quad (2)$$

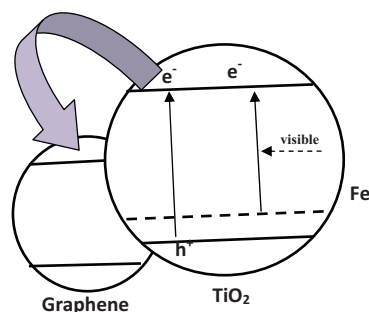
As shown in Fig. 8-inset, the photocatalytic degradation followed pseudo-first-order reaction kinetics in the studied concentration range. The half-life of E2 can then be calculated as per Eq. (3):

$$t_{1/2} = \frac{\ln 2}{k_{app}} \quad (3)$$

The calculated half-life of E2 at 1 sun intensity for 0.6% Fe doped TiO₂ nano wires/graphene sheet was only 41 min, whereas it was about 10 h in the absence of catalyst due to direct photolysis under 1 sun solar irradiation (AM 1.5G) using similar experimental conditions [1,2].

The enhancement in photocatalytic performance of the Fe-doped TiO₂/graphene samples can be explained by the higher surface area with mesoporous structure, lowered band gap and defects induced by Fe doping [47,48]. Although the measured band gap energy of both the 0.6% and 0.8% Fe doped TiO₂ nanowire/graphene assemblies was 2.25 eV, the surface area of the 0.8% Fe was 297 m²/g compared to 256 m²/g for the 0.6% Fe catalyst. The higher surface areas with mesoporous structure favours the adsorption of reactant molecules, as well as light absorbance. Fig. 7 inset plots k versus the BET surface area of the nanoassemblies showing a plateau in the rate with surface area. In addition, the increase of defects which were observed by the HRTEM images could capture more photoelectrons and thus reduce their recombination with photo induced holes. These factors would enhance the quantum efficiency, leading to the observed higher photocatalytic activity.

A proposed mechanism for charge transfer in the Fe doped TiO₂/FGs composites is shown in Scheme 1 in which the Fe doped TiO₂ nanoparticles are in intimate contact with FGs. The FGs have a work function around 4.2–4.5 eV [49], in which excited electrons from the Fe doped TiO₂ anatase conduction band can transfer to its (FGs) conduction band, resulting in narrowing the band gap, reduction in photoluminescence intensity, charge separation, stabilization, and hindering charge recombination. Moreover, Fe doped TiO₂/graphene photocatalysts can absorb more visible light which is favoured for photocatalytic reactions. In addition, when Fe doped TiO₂ were grown on the surface of graphene sheets, higher surface area photocatalysts were obtained; pollutant (E2) molecules were trapped on graphene pores, then Fe doped TiO₂ degrade them more efficiently in the visible area.



Scheme 1. Schematic of a proposed model for Fe doped TiO₂/FGSs with enhanced visible light photocatalytic reactivity.

4. Conclusions

Fe doped TiO₂ nanomaterials on the surface of graphene sheets were successfully synthesized by a simple sol–gel process using the green solvent, scCO₂. Fe doped TiO₂ nanowires were uniformly formed on the surface of graphene sheets with increasing surface area confirmed by BET analysis. UV–Vis analysis was used to show a reduction in the band gap of Fe doped TiO₂/graphene sheets compared to commercial TiO₂. By increasing the amount of Fe, the band gap decreased to the visible region with 0.6% Fe/TiO₂ being optimum, similarly confirmed by PL spectra. Photocatalytic activity of synthesized catalysts was successfully examined by the photodegradation of model environmental endocrine disruptors (E2) under solar visible irradiation. This work suggests that high porous, visible-light photocatalysis has tremendous potential for the development of an environmentally sustainable treatment process using sunlight in place of an artificial light source, since visible light comprises a much larger portion of the solar spectrum than UV light.

Acknowledgments

This work was financially supported by the Canadian Natural Science and Engineering Research Council (NSERC) and the Canadian Foundation for Innovation (CFI).

Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at [doi:10.1016/j.apcatb.2011.08.012](https://doi.org/10.1016/j.apcatb.2011.08.012).

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